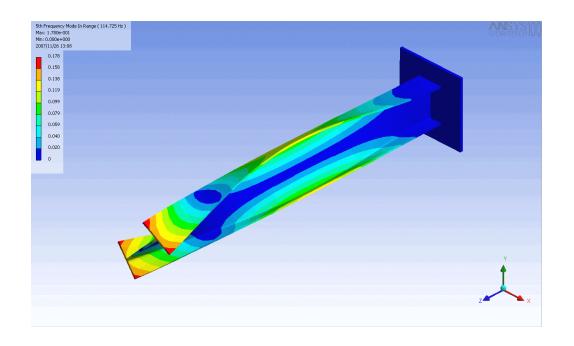
#### Operator learning for history-dependent and multiscale problems

Margaret Trautner

Data Assimilation and Inverse Problems for Digital Twins

October 8, 2025

#### Motivation via example: predicting material dynamics



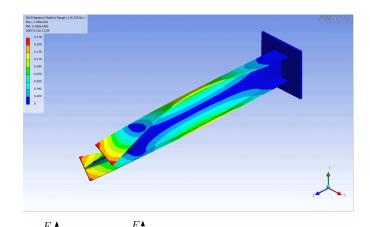


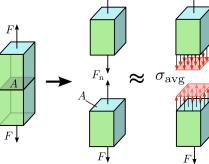
#### Constitutive laws

Equations for material dynamics:

- 1.  $F = m \cdot a$
- 2. Initial condition
- 3. Boundary condition
- 4. Constitutive law: relates material displacement to force (material strain to material stress)

Stress (normal): force per area Strain: gradient of displacement





# Simple constitutive law: linear elasticity

Mass (density)

Acceleration

Force

Force balance

 $\rho \partial_t^2 u(x,t) = \nabla \cdot \sigma(x,t) + f(x,t)$ 

Constitutive law (linear elasticity)

 $\sigma(x,t) = A \nabla_x u(x,t)$ 



Zero initial conditions

Zero Dirichlet boundary conditions

#### Multiscale constitutive law

Displacement

Stress

External forcing

Force balance

$$\rho \partial_t^2 u_{\varepsilon}(x,t) = \nabla_x \cdot \sigma_{\varepsilon}(x,t) + f(x,t),$$

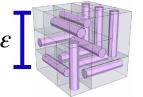
Constitutive law

$$\sigma_{\varepsilon}(x,t) = \Psi_{\varepsilon}^{\dagger} (\{\nabla_x u_{\varepsilon}(x,s)\}_{s \in [0,t]}, M)(t)$$

Zero initial conditions

Strain trajectory

Zero Dirichlet boundary conditions

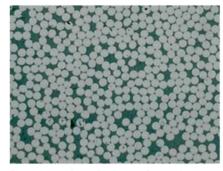


 $\varepsilon$  indicates dependence on small scale variable  $y = \frac{x}{\varepsilon}$ 

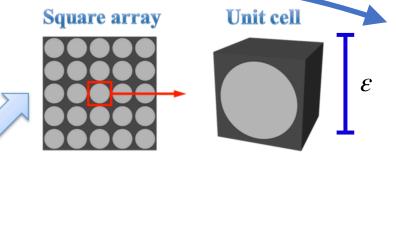


# Approaching multiscale problems

1. Assume periodicity in  $\varepsilon$ 



Cross-sectional view of continuous fiber reinforced composites



2. Use known physics to solve PDE on unit cell

3. Use resultant stress on all repeated cells to find displacement at next time iteration

# Homogenized equations

Displacement Stress External forcing  $\rho \partial_t^2 u_0(x,t) = \nabla_x \cdot \sigma_0(x,t) + f(x,t),$   $\sigma_0(x,t) = \Psi_0^\dagger \big( \{ \nabla_x u_0(x,s) \}_{s \in [0,t]}; M \big)(t)$  ions

Zero initial conditions

Force balance

Constitutive law

Zero Dirichlet boundary conditions

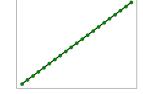
- Assume microstructure  $\varepsilon$  periodic
- Seek homogenized constitutive law  $\Psi_0^\dagger$  such that  $u_\varepsilon \to u_0$  as  $\varepsilon \to 0$
- $\Psi_0^{\dagger}$ : homogenized strain  $\mapsto$  homogenized stress
- Homogenized constitutive law maps between functions in time

# Operator learning

- Classical machine learning methods map between finite-dimensional spaces
- PDEs map between function spaces
- Operator learning approximates infinite-dimensional operators using data

Example: Learn derivative map  $\frac{d}{dx}$ :  $C^1([0,1]) \mapsto C([0,1])$ 



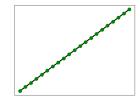


	Learning in Finite Dimensions	Learning in Infinite Dimensions
Mapping between	$\mathbb{R}^{21} \mapsto \mathbb{R}^{21}$	$C^1([0,1]) \mapsto C([0,1])$
Discretization	Hardwired into model architecture	Affects learned model parameters only indirectly
Evaluation at new	Can only be interpolated from model	Model itself can be evaluated at new
output points	output points	points

# When is operator learning useful?

- Classical methods work well to solve PDEs
- Operator learning only potentially useful if:
- 1. Data generation/training cost can be amortized
- 2. Equations are unknown





	Learning in Finite Dimensions	Learning in Infinite Dimensions
Mapping between	$\mathbb{R}^{21} \mapsto \mathbb{R}^{21}$	$C^1([0,1]) \mapsto C([0,1])$
Discretization	Hardwired into model architecture	Affects learned model parameters only indirectly
Evaluation at new output points	Can only be interpolated from model output points	Model itself can be evaluated at new points

#### Some existing work on operator learning

#### I. Architectures

• Fourier Neural Operators (FNO) (Li et al. 2021), DeepONet (Lu et al. 2021), GNO (Li et al. 2020), PCA-Net (Liu et al. 2022), ANO (Lanthaler et al. 2023), etc.

#### II. Analysis

- Universal approximation: FNO (Kovachki et al. 2021), DeepONet (Lu et al. 2021), etc. UA in finite dimensions goes back to Cybenko (1989).
- Error bounds with respect to model size: FNO (Kovachki et al. 2021), DeepONet (Lanthaler et al. 2021), DON with Lipschitz/Holder maps (Schwab et al. 2023) etc. Foundational work in finite dimensions by Yarotsky (2017).

#### III. Application setting of constitutive modeling in solid mechanics

• PCA-Net approach to multiscale setting (Liu et al. 2022), deep learning approach to constitutive models (Huang et al. 2020) (Mozaffar et al. 2019), structure-preserving deep learning approach to constitutive models (As'ad et al. 2022), etc.



Outline: Operator learning for history-dependent and multiscale problems

I. Recurrent neural operator for causal history-dependent models

II. Learning for discontinuous inputs (materials) with FNO

III. Comparison of parameter-to-observable operator learning for multiscale elasticity via Fourier Neural Mapping (FNM)

IV. Sampling convergence rate theory-to-practice gap

# History-dependent constitutive laws

- For some materials, stress depend on entire strain history
- Kelvin-Voigt viscoelastic material



$$\sigma_{\varepsilon}(t) = E_{\varepsilon} \operatorname{strain}_{\varepsilon}(t) + \nu_{\varepsilon} \partial_{t} \operatorname{strain}_{\varepsilon}(t)$$

Homogenized:

$$\sigma(t) = E'\operatorname{strain}(t) + \nu' \partial_t \operatorname{strain}(t) - \int_0^t \kappa(t - \tau) \operatorname{strain}(\tau) \, d\tau$$

### Markovian model to capture history dependence

#### Recurrent neural operator (RNO) architecture

$$\sigma(t) = F(\operatorname{strain}(t), \partial_t \operatorname{strain}(t), \xi(t))$$

$$\dot{\xi}(t) = G(\operatorname{strain}(t), \xi(t))$$

$$\xi(0) = 0$$

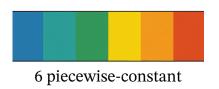
- $\xi \in \mathbb{R}^L$  are hidden variables
- F and G are neural networks
- $\xi$  updated with any time integration scheme

- $\xi$  captures memory
- Discretization invariant

# RNO model and 1d viscoelasticity

• For piecewise-constant viscoelastic materials, homogenized constitutive law takes the form

$$\begin{split} &\sigma(t) = E' \mathrm{strain}(t) + \nu' \partial_t \mathrm{strain}(t) - \left\langle \xi(t), \mathbf{1}_L \right\rangle \\ &\dot{\xi}(t) = \beta \mathrm{strain}(t) - \alpha \xi(t) \\ &\xi(0) = 0 \end{split}$$



• Exactly matches RNO model architecture

# RNO architecture $\sigma(t) = F(\operatorname{strain}(t), \partial_t \operatorname{strain}(t), \xi(t))$ $\dot{\xi}(t) = G(\operatorname{strain}(t), \xi(t))$ $\xi(0) = 0$



## RNO approximation theorem

#### RNO Approximation Theorem (Informal)

Let  $\Psi_0^{PC}$  be the Markovian constitutive law for 1d piecewise-constant viscoelastic materials.

For any  $\eta > 0$ , there exists an RNO  $\Psi_0^{\text{RNO}}$  such that

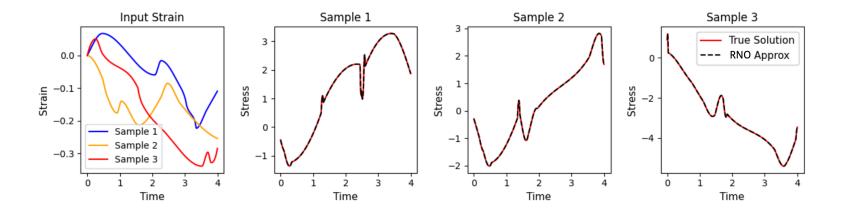
$$\sup_{t\in[0,T]}\left|\Psi_0^{\text{PC}}(\{\operatorname{strain}(\tau)\}_{\tau\in[0,T]},t)-\Psi_0^{\text{RNO}}(\{\operatorname{strain}(\tau)\}_{\tau\in[0,T]},t)\right|<\eta,$$

• Also show that  $u_{\varepsilon}$  for piecewise-continuous materials is approximated by  $u_0^{PC}$  for some choice of piecewise-constant materials

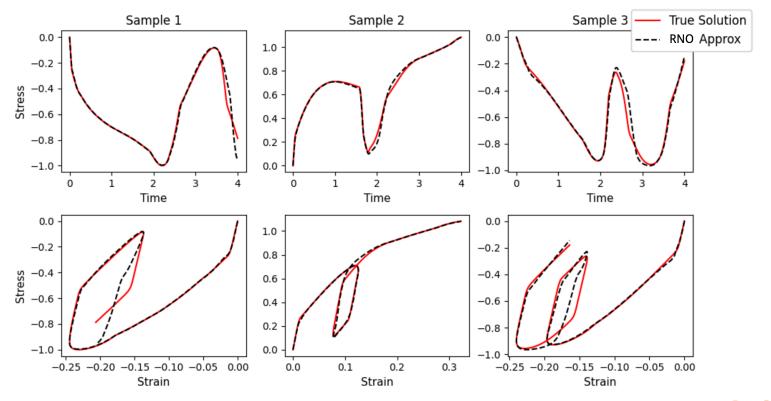


RNO is proven to approximate the history-dependent constitutive law

#### Experimental results: 1d piecewise-constant viscoelasticity

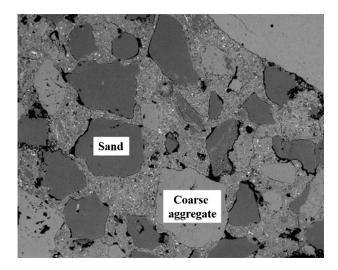


# Experimental results: 1d elasto-viscoplasticity

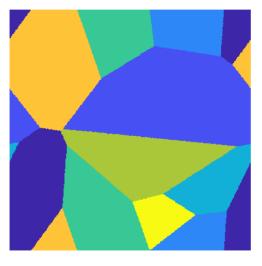




#### Material models in 2d



Grains in a cement-based material



Voronoi tessellation

Learn 2d elasticity constitutive model using a Fourier Neural Operator

# Fourier Neural Operator (FNO)



FNO  $\Psi$  is an integral kernel based neural operator architecture mapping between Banach spaces  $\mathcal{A} \to \mathcal{U}$ 

FNO Model

$$\Psi = \mathcal{Q} \circ \mathsf{L}_{T-1} \circ \dots \circ \mathsf{L}_0 \circ \mathscr{P}$$

Hidden FNO Layers

$$\mathsf{L}_t v_t = \sigma_t (W_t v_t + \mathscr{K}_t v_t + b_t)$$

Integral Kernel Operator 
$$(\mathcal{K}_t v_t)(x) = \sum_{k \in \mathbb{Z}^d} (P_t^{(k)}) \hat{v}_t(k) e^{2\pi i \langle k, x \rangle}$$

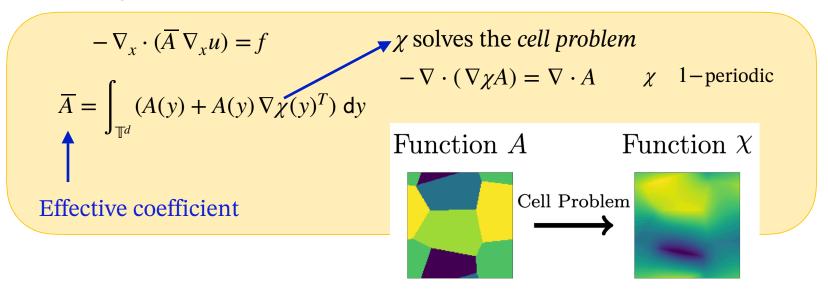
Fourier coefficients of  $v_t$ 

# Linear multiscale elasticity

$$-\nabla_{x}\cdot(A_{\varepsilon}\nabla_{x}u_{\varepsilon})=f$$

$$A_{\varepsilon}(x) = A\left(\frac{x}{\varepsilon}\right)$$
  $A(\cdot)$  1-periodic

#### Homogenized equations:



Interested in map  $A \mapsto \chi$ .

# Stability and universal approximation

Goal: approximate  $A \mapsto \chi$  using FNO.

Universal approximation requires

- (1) a separable input space and
- (2) a continuous true map.

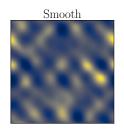
$A \mapsto \chi$	Separable Input Space?	Continuous map?
$L^{\infty} \mapsto \dot{H}^1$		
$L^p \mapsto \dot{H}^1, p \in [2, \infty)$		Our result

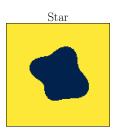
# Continuity result

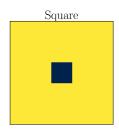
#### Cell Problem Continuity (Informal)

- $A \mapsto \chi$  is *continuous* from closed set in  $L^2$  to  $\dot{H}^1$
- There exists  $q_0>2$  such that for all  $q\in (q_0,\infty), A\mapsto \chi$  is Lipschitz continuous from  $L^q$  to  $\dot{H}^1$

Continuity result + compactness of input set = Universal Approximation Input set  $\subset$  BV  $\cap$   $L^{\infty}$  is sufficient for compactness in  $L^2$ .



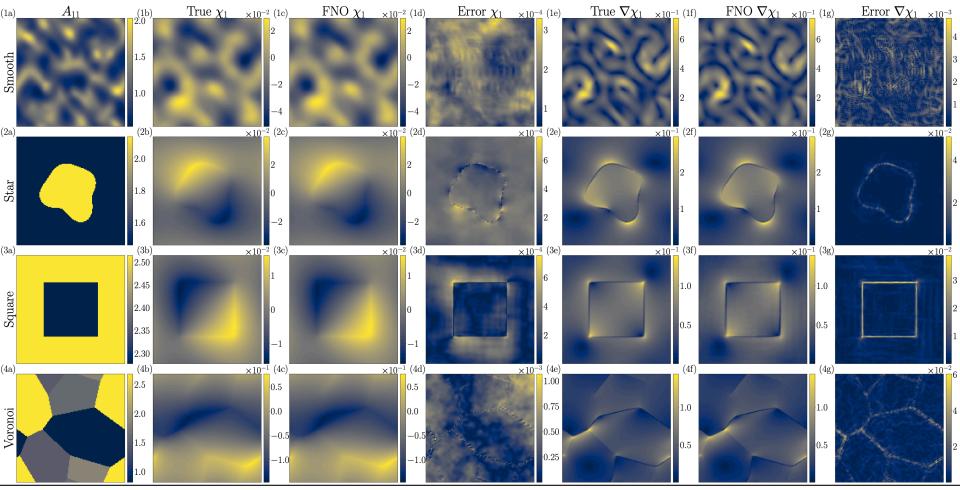




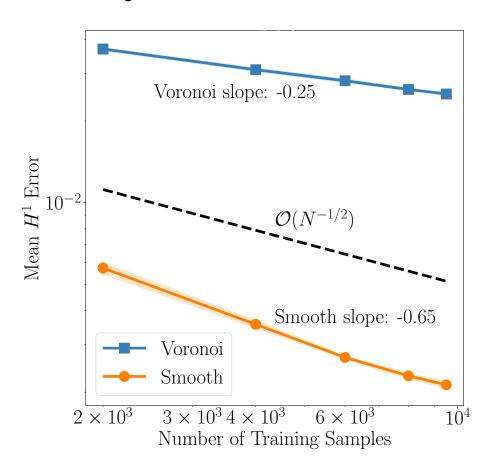




Operator learning is mathematically justified for a variety of common microstructures

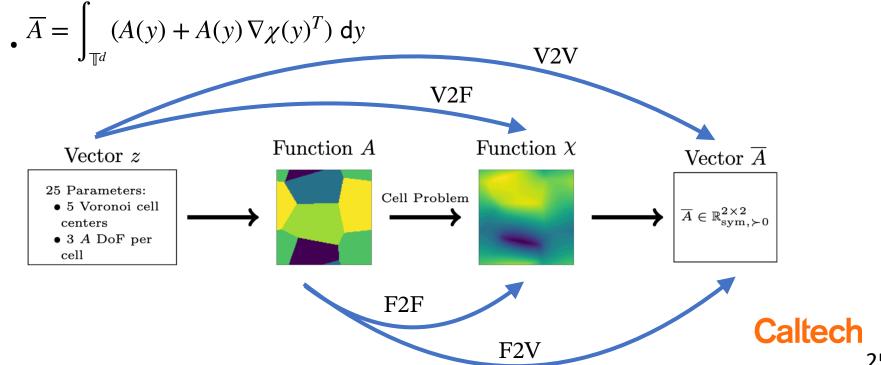


# Data efficiency



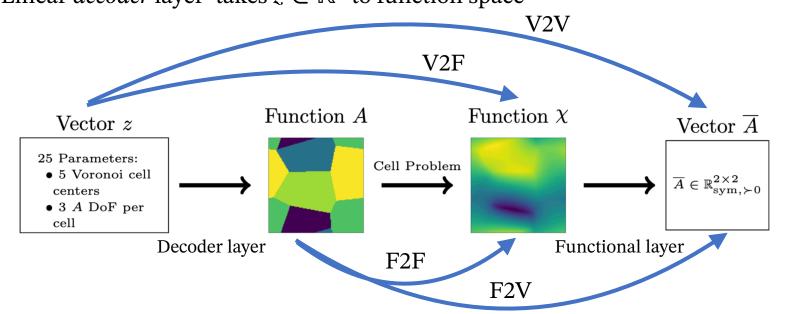
## Different ways to learn effective coefficient $\overline{A}$

- Learned  $A \mapsto \chi$ , a map between function spaces
- Would it be more data efficient to learn effective coefficient directly, i.e.  $A \mapsto A$ ?

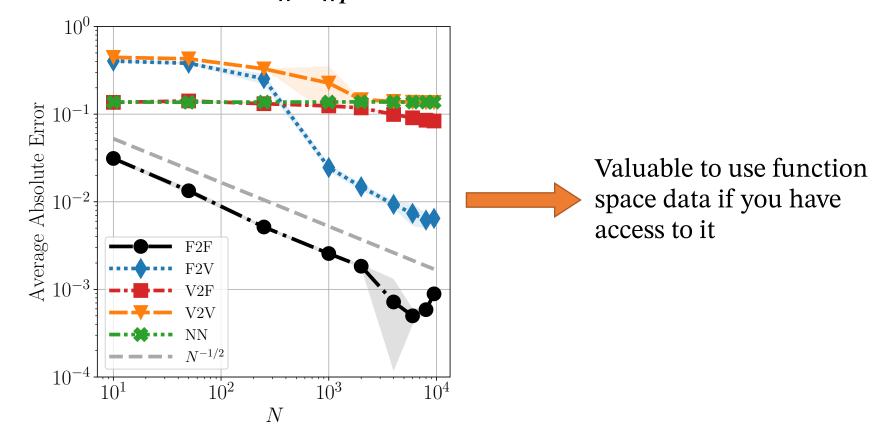


#### Fourier Neural Mappings: "parameter-to-observable"

Modification of FNO to accommodate finite-dimensional inputs and outputs Linear *functional* layer takes functions to  $\mathbb{R}^n$  Linear *decoder* layer takes  $z \in \mathbb{R}^n$  to function space



# Absolute error in $\|\overline{A}\|_F$ versus data size N

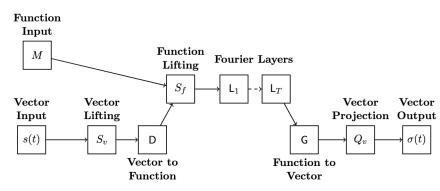


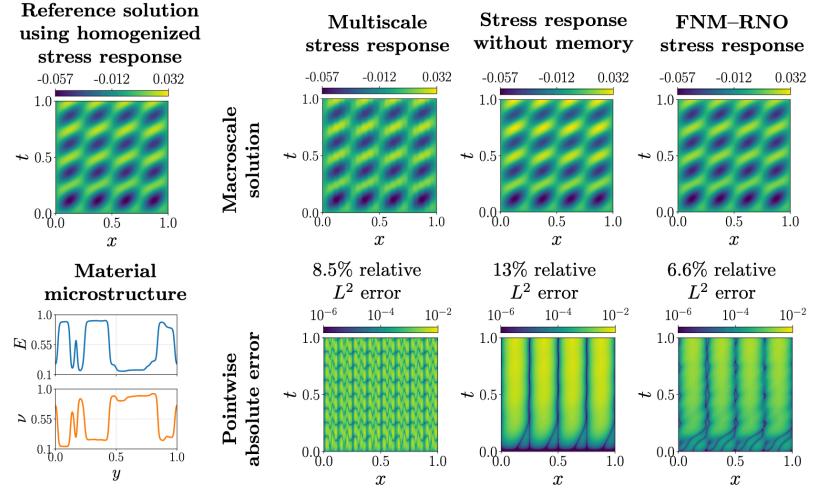
#### Learning material dependence and memory simultaneously

# RNO architecture $\sigma(t) = F(\overline{e}(t), \dot{\overline{e}}(t), \xi(t))$ $\dot{\xi}(t) = G(\overline{e}(t), \xi(t))$ Fourier Neural Mappings Mappings



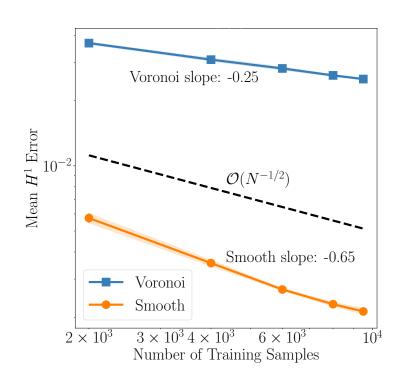
The same model can learn history-dependence and materialdependence simultaneously.

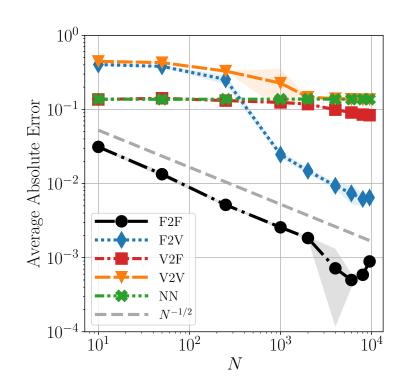




Bhattacharya, Kaushik, Lianghao Cao, George Stepaniants, Andrew Stuart, and **M. Trautner**. "Learning Memory and Material Dependent Constitutive Laws." *arXiv preprint* arXiv:2502.05463 (2025).

#### Sampling convergence rates- what does theory tell us?







### Theory-to-practice gap: introduction

#### Theory:

- Universal approximation: model expressivity
- Model size bounds: how wide and deep a model needs to be to achieve low error

#### Practice:

• Must identify model from finite data: data efficiency

Is there a relationship between models with high model expressivity and models with high data efficiency?

#### Error convergence rates

Theory: parametric convergence rate

• A model with *n* parameters can approximate with error  $\lesssim n^{-\alpha}$ 

Practice: sampling convergence rate

• A model given N data samples can find a reconstruction with error  $\lesssim N^{-\beta_*}$ 

High  $\alpha$  = good model expressivity

High  $\beta_* = \text{good data efficiency}$ 

What is the relationship between  $\alpha$  and  $\beta_*$ ?

## Convergence rate "gap" for neural networks

High  $\alpha$  = good model expressivity High  $\beta_*$  = good data efficiency

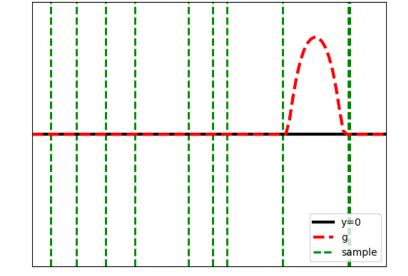
Degrees of freedom argument suggests  $\beta_* = \alpha$ For polynomial reconstruction, determining n parameters requires N = n + 1 samples.

For neural networks, there is a gap between  $\alpha$  and  $\beta_*$ .



#### Theory-to-practice gap in finite dimensions: intuition

- For any sample points  $x_1, ..., x_N$ , there exists a sufficiently large "void"
- A locally supported function *g* in this void is indistinguishable from the zero function
- ReLU neural networks can efficiently approximate certain locally supported functions





# Theory-to-practice gap in finite dimensions

 $\alpha$  is the *parametric convergence rate*: error in the number of nonzero weights n converges at a rate  $n^{-\alpha}$ . High  $\alpha =$  good model expressivity.

 $\beta_*$  is the *sampling convergence rate* for approximation in  $L^p$ : expected  $L^p$  error of the reconstruction with N data samples converges at a rate  $N^{-\beta_*}$ . High  $\beta = \text{good data efficiency}$ .

Theorem (Theory-to-practice gap in finite dimensions) (Informal)

For approximation of functions  $[0,1]^d \mapsto \mathbb{R}$  with parametric convergence rate  $\alpha$ ,

$$\beta_* \le \frac{1}{p} + \frac{1}{d} \cdot \frac{\alpha}{\alpha + \text{correction}^*}.$$

## Theory-to-practice gap in infinite dimensions

 $\alpha$  is the *parametric convergence rate*: error in the number of nonzero weights n converges at a rate  $n^{-\alpha}$ . High  $\alpha =$  good model expressivity.

 $\beta_*$  is the *sampling convergence rate* for approximation in  $L^p$ : expected  $L^p$  error of the reconstruction with N data samples converges at a rate  $N^{-\beta_*}$ . High  $\beta = \text{good data efficiency}$ .

Theorem (Theory-to-practice gap in infinite dimensions) (Informal) For approximation of functions  $\mathcal{X} \mapsto \mathbb{R}$  with parametric convergence rate  $\alpha$ ,

$$\beta_* \leq \frac{1}{p}$$
.

### Conclusion

- I. Recurrent neural operator captures history dependence
  - Effective in various application settings in multiscale constitutive surrogate modeling.
- II. In 2d, we prove that learning on discontinuous materials is theoretically valid
  - Numerical experiments show that FNO can be empirically effective in this setting.
  - Bound discretization error of FNO due to aliasing as well.
- III. Modified the FNO to accept finite-dimensional inputs and outputs
  - Data efficiency of learning in this setting is explored.
  - This modification also allows for material dependence to be built into the model.
- IV. Hardness result for sampling convergence rate via the lens of the "theory-to-practice gap"

Appendix

### Summary of operator learning for constitutive models

- I. Recurrent neural operator captures history dependence
  - For 1d viscoelasticity, use is rigorously justified.
  - Effective in setting of 1d elasto-viscoplasticity, more complex materials as well.
- II. In 2d, we prove that learning on discontinuous materials is theoretically valid
  - Numerical experiments show that FNO can be empirically effective in this setting.
- III. Modified the FNO to accept finite-dimensional inputs and outputs
  - Data efficiency of learning in this setting is explored.
  - This modification also allows for material dependence to be built into the model.



# Discretization error in FNO (aliasing error)

In definition, 
$$\widehat{v}_t(k) = \int_{\mathbb{T}^d} v_t(x)e^{-2\pi i \langle k, x \rangle} dx$$

In practice:  $\hat{v}_t^N(k) = \mathsf{DFT}(v_t^N)$ 

$$\Psi = \mathcal{Q} \circ L_{T-1} \circ \dots \circ L_0 \circ \mathcal{P}$$
 
$$L_t v_t = \sigma_t (W_t v_t + \mathcal{K}_t v_t + b_t)$$
 
$$(\mathcal{K}_t v_t)(x) = \sum_{k \in \mathbb{Z}^d} (P_t^{(k)}) \widehat{v}_t(k) e^{2\pi i \langle k, x \rangle}$$

#### Theorem (Informal) (Bounding FNO discretization error)

For input a with Sobolev regularity s, the error evaluated on the grid  $[N]^d$  of the output of layer  $L_t$  satisfies

$$\frac{1}{N^{d/2}} \| v_t(a) - v_t^N(a) \|_{\ell^2([N]^d)} \le CN^{-s}$$

# Setting: multiscale material response

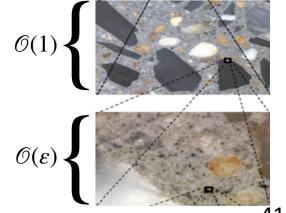
Force balance Displacement 
$$\int_t^{\text{Displacement}} \nabla u_{\varepsilon}(x,t) = \nabla_x \cdot \sigma_{\varepsilon}(x,t) + f(x,t),$$

$$\sigma_{\varepsilon}(x,t) = \Psi_{\varepsilon}^{\dagger} (\{ \nabla_x u_{\varepsilon}(x,s) \}_{s \in [0,t]}; M, x)(t),$$

$$u_{\varepsilon}(x,0) = \partial_t u_{\varepsilon}(x,0) = 0$$
, Strain history

 $u_{\varepsilon}(x,t) = 0, \quad x \in \partial \mathcal{D}.$ 

$$\varepsilon$$
 indicates dependence on small scale variable  $y = \frac{x}{-}$ 



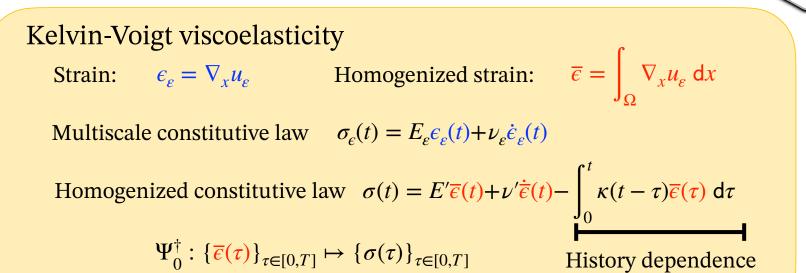
# Homogenized equations

Force balance 
$$\rho \partial_t^2 u_0(x,t) = \nabla_x \cdot \sigma_0(x,t) + f(x,t),$$
 Constitutive law 
$$\sigma_0(x,t) = \Psi_0^\dagger \big( \{ \nabla_x u_0(x,s) \}_{s \in [0,t]}; M \big)(t),$$
 Initial condition 
$$u_0(x,0) = \partial_t u_0(x,0) = 0, \quad \text{Strain history}$$
 Boundary condition 
$$u_0(x,t) = 0, \quad x \in \partial \mathcal{D}.$$

- Assume microstructure  $\varepsilon$  periodic
- Seek homogenized constitutive law  $\Psi_0^{\dagger}$  such that  $u_{\varepsilon} \to u_0$  as  $\varepsilon \to 0$
- $\Psi_0^{\dagger}$ : homogenized strain  $\mapsto$  homogenized stress

### History-dependent constitutive laws

• For some materials, stress depend on entire strain history



### RNO approximation theorem

#### RNO Approximation Theorem (Informal)

Let  $\Psi_0^{PC}$  be the Markovian form taken by  $\Psi_0^{\dagger}$  for 1d PC viscoelastic materials.

For any  $\eta > 0$ , there exists  $\Psi_0^{RNO}$  such that

$$\sup_{t \in [0,T]; \epsilon \in \mathsf{Z}} \left| \Psi_0^{\mathsf{PC}}(\{\epsilon(\tau)\}_{\tau \in [0,T]}, t) - \Psi_0^{\mathsf{RNO}}(\{\epsilon(\tau)\}_{\tau \in [0,T]}, t) \right| < \eta,$$

where Z bounds  $\epsilon$  uniformly in t.

- Also show that  $u_{\varepsilon}$  for piecewise-continuous materials is approximated by  $u_0^{PC}$  for some choice of PC materials
- Francfort & Suquet (1986) gives  $u_{\varepsilon} \rightarrow u_0$



#### Homogenization two ways

- 1. We have defined homogenized map  $\Psi_0^{\dagger}$ :  $\{\overline{\epsilon}(\tau)\}_{0 \leq \tau \leq t} \mapsto \{\sigma(\tau)\}_{0 \leq \tau \leq t}$  where  $\overline{\epsilon} = \nabla_x u_0$  as the map from the homogenized strain trajectory to the homogenized stress trajectory
- 2. Another perspective views  $\Psi_0^{\dagger}$  as the map from spatially-averaged strain trajectory over the cell to spatially averaged stress trajectory over the cell, i.e.

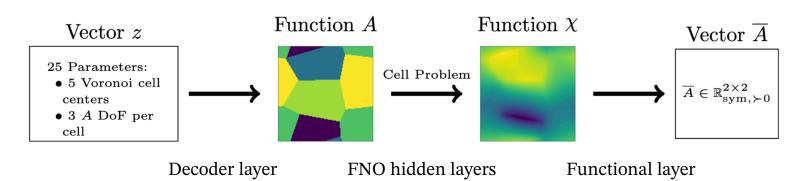
$$\int_{\mathbb{T}^d} \nabla_x u_{\varepsilon}(y) \, dy \text{ to } \int_{\mathbb{T}^d} \sigma_{\varepsilon}(y) \, dy.$$

### Fourier Neural Mappings

• Modification of FNO to accommodate finite-dimensional inputs and outputs with underlying infinite-dimensional operator map

Linear *functional* layer 
$$h \mapsto \mathcal{G}h := \int_{\Omega} \kappa(x)h(x) \, dx$$
 takes functions to  $\mathbb{R}^n$ 

Linear decoder layer  $z \mapsto \mathcal{D}z := \kappa(\cdot)z$  takes  $z \in \mathbb{R}^n$  to function space



### Theory-to-practice gap: introduction

Many theoretical results in machine learning answer questions about *model expressivity*:

- Can this model class approximate to  $\varepsilon$  accuracy any function in a certain set? (*Universal approximation*)
- Given *n* nonzero model weights, how small is the error of the optimal model parameterization? (*Parametric complexity*)

Other results answer questions about *generalizability*:

- Given *N* data samples from the underlying map of interest, how well does Algorithm X do, on average, compared to the optimal model parameterization? (*Optimization*)
- Given *N* data samples, how well can *any* algorithm reconstruct the true underlying map of interest? (*Sampling complexity*)



# Infinite-dimensional setting

	Finite-Dimensional Setting	Infinite-Dimensional Setting
Mapping between	$\mathbb{R}^d \to \mathbb{R}$	$\mathcal{X} \to \mathbb{R}$ for separable Banach space $\mathcal{X}$
Restriction of input space	$D = [0,1]^d \subset \mathbb{R}^d$	Measure on $\mathcal X$
Expected error measured in	$L^p(D)$	Banach space $V$ such that $U \subset V$

### More formal set-up: Neural Operator Approximation Spaces

- Approximation Space  $A^{\alpha} = \{ \mathcal{G} \in C(\mathcal{X}; \mathcal{Y}) : ||\mathcal{G}||_{A^{\alpha}} < \infty \}$
- Approximation space quasi-norm  $\|\mathcal{G}\|_{A^{\alpha}}=\inf\{\theta>0:\Gamma^{\alpha}(\mathcal{G}/\theta)\leq 1\}$

$$\Gamma^{\alpha}(\mathcal{G}) = \max\{ \sup_{u \in \mathcal{X}} \|\mathcal{G}(u)\|, \sup_{n \in \mathbb{N}} [n^{\alpha} \cdot \inf_{\Psi \in \Sigma_{n}} \sup_{u \in \mathcal{X}} \|\mathcal{G}(u) - \Psi(u)\|_{\mathcal{Y}}] \}$$

•  $\Sigma_n$  contains neural operators with at most n nonzero weights of uniformly bounded magnitude

 $A^{\alpha}$  contains all functions  $\mathcal{X} \to \mathcal{Y}$  that can be uniformly approximated at rate  $n^{-\alpha}$  by neural operators with at most n nonzero weights (parameters) of bounded magnitude

• We work in unit ball  $U^{\alpha} \subset A^{\alpha}$ 

### More formal set-up: Algorithms

- $Alg_N(U, V)$  set of all deterministic methods using N point measurements
- $\mathscr{A} \in \operatorname{Alg}_N(U, V)$  if there exist samples  $(x_1, ..., x_N) \in \mathscr{X}^N$  and map  $Q: \mathbb{R}^N \to V$  such that  $\mathscr{A}(G) = Q(G(x_1), ..., G(x_N))$  for all  $G \in U$
- Optimal error  $e_N = \inf_{\mathscr{A} \in Alg_{_N}} \sup_{G \in U} \|\mathscr{A}(G) G\|_V$
- Can also formulate set of randomized algorithms
- Results hold over all reconstruction methods using *N* samples

#### Proof sketch

- 1. Show that approximation space  $U \subset C(\mathcal{X}, \mathbb{R})$  contains compositions of the form  $f \circ \mathcal{E}$ , where  $\mathcal{E} : \mathcal{X} \to \mathbb{R}^d$  is an encoder, and  $f : \mathbb{R}^d \to \mathbb{R}$  belongs to the d-dimensional approximation space  $f \in U_{\ell}^{\alpha,\infty}$ .
- 2. Then  $f \mapsto f \circ \mathscr{E}$  defines an embedding of  $U_{\ell}^{\alpha,\infty}$  into U, so any d-dimensional bound on the convergence rate applies in the infinite-dimensional setting.

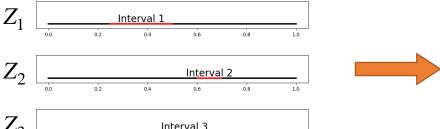
#### Proof sketch

Require measure  $\mu$  on  $\mathcal{X}$  and encoder  $\mathcal{E}: \mathcal{X} \mapsto \mathbb{R}^d$  such that  $\mathcal{E}$  "fills out" the unit cube  $[0,1]^d$  in the following sense:

$$\mathscr{E}_{\#}\mu \geq c \cdot \text{Unif}([0,1]^d)$$

Assume  $\mu \in \mathcal{P}(\mathcal{X})$  can be written as the law of  $u = \sum_{j=1}^{\infty} Z_j e_j$  for biorthogonal sequences

 $\{e_j\}_{j\in\mathbb{N}}\subset\mathcal{X},\,\{e_j^*\}_{j\in\mathbb{N}}\subset\mathcal{X}^*$  and iid real-valued random variables  $Z_j$  with sufficient density on open intervals.



By approximating a projection onto  $\{e_j\}_{j\in[d]}$ , the encoder "fills out" a cube with sufficient density that can be rescaled to  $[0,1]^d$ 

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